

Preparation of hafnium oxide thin films by sol–gel method

Z. J. Wang · T. Kumagai · H. Kokawa ·
M. Ichiki · R. Maeda

Published online: 13 September 2007
© Springer Science + Business Media, LLC 2007

Abstract Hafnium oxide (HfO₂) films were grown on SiO₂/Si substrates by a sol–gel method, and their crystalline structure, microstructure and electrical properties were investigated. X-ray diffraction analysis indicated that the monoclinic HfO₂ films could be obtained by annealing at 500 °C. A transmission electron microscopy (TEM) image showed that the films were grown as a spherulite grain structure with a mean grain size of approximately 15 nm. The dielectric constant of the HfO₂ films of 300 nm was approximately 21.6, and the current–voltage measurements showed that the leakage current density of the HfO₂ films was approximately 1.14×10^{-5} A/cm² at an applied electric field of 100 kV/cm. The sol–gel method-fabricated HfO₂ films are concluded to be feasible for MEMS applications, such as capacitive-type MEMS switches.

Keywords Hafnium oxide · Thin film · Crystalline structure · Microstructure · Electrical properties

1 Introduction

Hafnium oxide (HfO₂) films have attracted great interest, because of their large number of potential applications in

optical coatings and microelectronics [1–5]. The large dielectric constant and good thermal stability when in contact with silicon, makes HfO₂ film a promising candidate to replace the current SiO₂ gate dielectric materials in field-effect devices [4, 5]. It can also be applied a dielectric in microelectro-mechanical systems (MEMS). A novel approach using a PZT/HfO₂ multilayered dielectric for capacitive-type MEMS switches was investigated in our previous work [6]. It is found that compared with Si₃N₄, the use of a PZT/HfO₂ multilayered dielectric enables the realization of a high equivalent dielectric constant of 79–82 and a low leakage current density after a bias stressing time of 4¹⁰ s, resulting in an efficient switching isolation and a very low power consumption. Many the techniques have been used to fabricate HfO₂ thin films, such as sputtering [7], the sol–gel method [8], atomic layer deposition [9], and metal–organic chemical vapor deposition (MOCVD) [10]. The sol–gel method generally offers significant advantages in the film fabrication of electronic materials, such as high purity, ease of composition control, relatively low processing temperature and large deposition area [11]. In this study, HfO₂ films were deposited by the sol–gel method on SiO₂/Si substrates and their crystalline phases, microstructures and electrical properties were investigated.

2 Experimental procedure

SiO₂/Si(100) substrates were prepared by thermally oxidizing (100) silicon wafers. The HfO₂ films were grown by the sol–gel method using a precursor solution prepared by the Kojundo Chemical Laboratory. The films were coated on SiO₂/Si substrates using a spin coater operated at 500 rpm for 10 s and 2,000 rpm for 10 s. The coated films were

Z. J. Wang (✉) · T. Kumagai · H. Kokawa
Department of Materials Processing,
Graduate School of Engineering, Tohoku University,
Aoba-yama 6-6-02,
Sendai 980-8579, Japan
e-mail: wangzj@argon.material.tohoku.ac.jp

M. Ichiki · R. Maeda
National Institute of Advanced Industrial Science and Technology,
1-2 Namiki,
Tsukuba 305-8564, Japan

dried at 120 °C for 5 min and 400 °C for 5 min, and finally crystallized at temperatures ranging from 500 to 750 °C for 30 min. For obtaining more reliable results of XRD analysis, the coating and heat-treatment process was repeated 11 times to obtain thicker HfO₂ films.

The crystal structure of the HfO₂ films was examined by X-ray diffraction (XRD, Rigaku RINT2000, CuK α radiation) analysis. The surface and cross-sectional morphologies of the films were observed by scanning electron microscopy (FE-SEM, JSM-6500F). Cross-sectional TEM specimens were prepared by conventional grinding and polishing. The specimens were ground and polished to a thickness of 20 μ m, and were further thinned to perforation by Ar-ion milling using a Gatan precision ion polishing system (PIPS). The microstructures of the films were studied by transmission electron microscopy (HF2000, HITACHI). For measuring the electrical properties of the films, a Pt/Ti film was used as the bottom electrode, and a Pt film (size= ϕ 1.5 mm) was deposited by sputtering to form the top electrode. The dielectric constant and current–voltage (I – V) characteristics of these HfO₂ films were measured using multifrequency LCR meter (HIOKI 3522-50).

3 Results and discussion

3.1 Crystalline phases of HfO₂ films

First, to obtain a more reliable estimation, the XRD analysis of the thick HfO₂ films with 11 coating cycles annealed at 750 °C was conducted. Figure 1 shows the XRD pattern of the HfO₂ films deposited on SiO₂/Si substrate. As can be

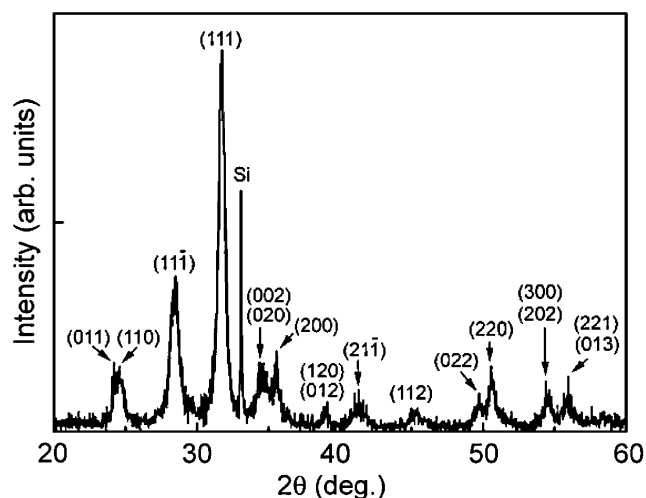


Fig. 1 XRD pattern of HfO₂ films with 11 coating cycles on SiO₂/Si(100) substrates annealed at 750 °C

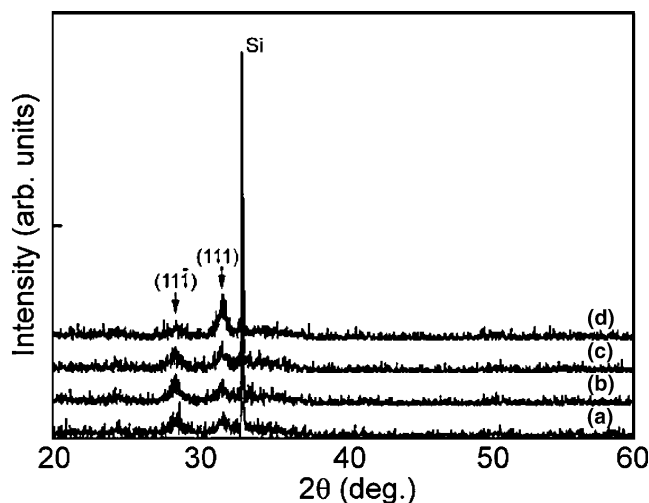


Fig. 2 XRD patterns of single-layer HfO₂ films on SiO₂/Si(100) substrates annealed at (a) 500 °C, (b) 550 °C, (c) 650 °C and (d) 750 °C

seen, distinct diffraction peaks can be observed, and these diffraction peaks are ascribed to the monoclinic phase of HfO₂ [12]. A peak ascribed to either tetragonal or orthorhombic HfO₂, often appearing at $2\theta=30.3^\circ$ in HfO₂ films deposited by atomic layer deposition and sputtering [13–16], was not detected in this study. The HfO₂ films, therefore, mainly crystallized in the monoclinic phase without the tetragonal or orthorhombic phase. There is no preferred orientation in these HfO₂ films, because the relative intensity ratios between all the diffraction peaks are close to those in the HfO₂ powder diffraction patterns [12]. Crystallite size (t) can be calculated using the Scherrer formula, $t=0.9 \lambda/(B \cos\theta)$, where λ is the X-ray wavelength, B is the peak width, and θ is the Bragg diffraction angle. The average crystallite size in the HfO₂ films calculated from the (111) and (11-1) diffraction peaks was approximately 15 nm.

Figure 2 contains representative results of the XRD analysis of single-layer HfO₂ films annealed at various temperatures on SiO₂/Si substrates. At an annealing temperature of 500 °C, the (111) and (11-1) peaks of monoclinic HfO₂ appear. The peak density of these peaks increases slightly with annealing temperature. These results reveal that monoclinic HfO₂ films can be obtained at the annealing temperature of 500 °C by the sol–gel method, which is significantly lower than that of the HfO₂ films deposited by sputtering [6]. In our previous study, the HfO₂ films were deposited at room temperature by sputtering and then crystallized by postdepositional annealing at a temperature of 600 °C. It is imperative to decrease processing temperature for HfO₂ films for microelectronics and MEMS applications.

3.2 Microstructure of HfO₂ films

Figure 3 shows a cross-sectional TEM image of the HfO₂ film with five coating cycles grown on SiO₂/Si substrate. The thickness of the films is approximately 275 nm. From the result, the thickness of one layer of the HfO₂ film is concluded to be approximately 55 nm. The films consist of densely packed grains with a mean grain size of approximately 15 nm. The grain size obtained by TEM is in very good agreement with that calculated by XRD analysis using the Scherrer formula, as described above. Figure 4 shows a high-resolution electron microscopy (HTEM) image of the HfO₂ films. The measured *d* spacing of the crystallite shown in Fig. 4 is 0.3162 nm, and is coincident with the (11-1) planes (0.315 nm) of the monoclinic phase of HfO₂. The measured *d* spacing of monoclinic-phase HfO₂ is also consistent with those calculated by XRD analysis.

3.3 Electrical properties of HfO₂ films

The dielectric constant measurement of the films was carried out at a frequency of 1 kHz using an impedance analyzer. Figure 5 shows the schematic of dielectric constants of HfO₂ films vs sweep frequency. The room temperature dielectric constant of HfO₂ films was 21.6, which is in good agreement with the data in the literature (22) [17] but slightly higher than that of sputter-deposited HfO₂ films (18–20) [5]. The high dielectric constant in this work could be attributed to the high quality of the HfO₂ films, which crystallized well in the monoclinic phase with densely packed grains. The *I*–*V* measurements showed that the leakage current density of the HfO₂ films was 1.14×10^{-5} A/cm² at an applied electric field of 100 kV/cm. From their electrical properties, the sol–gel method-fabricated

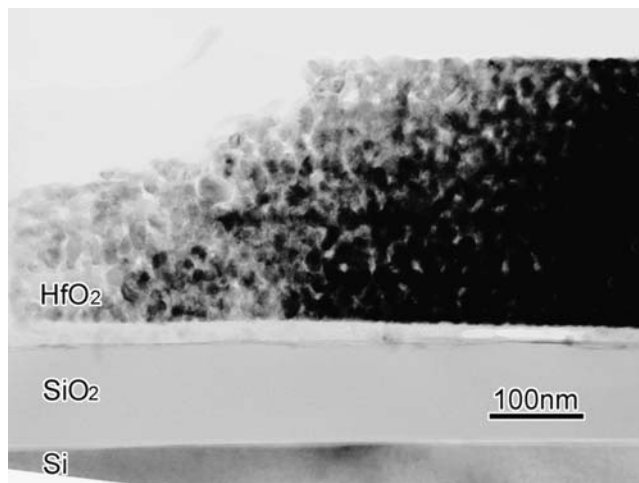


Fig. 3 Cross-sectional TEM image of HfO₂ films with five coating cycles deposited on SiO₂/Si(100) substrates

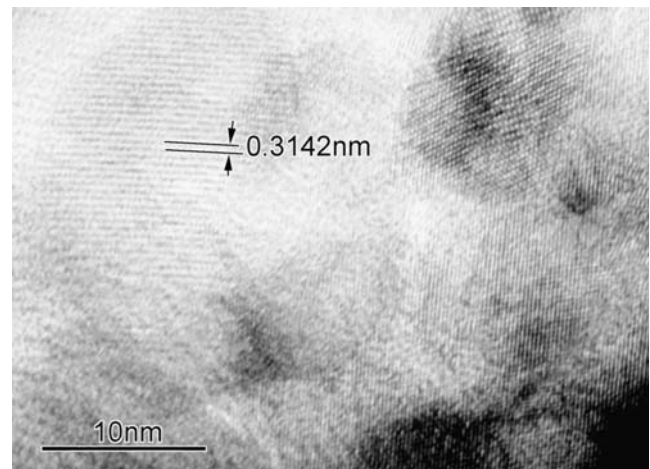


Fig. 4 Cross-sectional HTEM image of HfO₂ films with five coating cycles deposited on SiO₂/Si(100) substrates

HfO₂ films are concluded to be feasible for MEMS applications, such as capacitive-type MEMS switches.

4 Conclusions

Hafnium oxide (HfO₂) films were fabricated on SiO₂/Si (100) substrates by a sol–gel method. Their crystalline structure and microstructures were studied by XRD analysis and transmission electron microscopy, respectively. XRD analysis indicated that the monoclinic HfO₂ films could be obtained by annealing at 500 °C on SiO₂/Si substrates. TEM images revealed that HfO₂ films consist of densely packed grains with a mean grain size of approximately 15 nm. The electrical properties of the HfO₂ films were investigated by dielectric constant and current–voltage measurements. From their electrical properties, the sol–gel

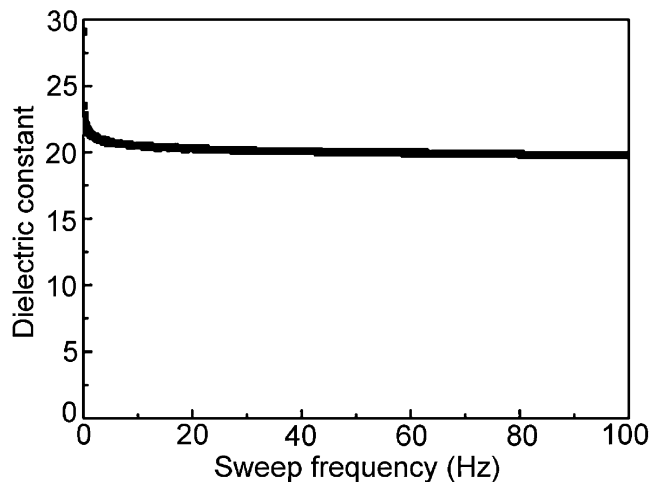


Fig. 5 Schematic of dielectric constants of HfO₂ films vs sweep frequency

method-fabricated HfO_2 films are concluded to be feasible for MEMS applications, such as capacitive-type MEMS switches.

Acknowledgement The support of this work by a grant-in-aid for Scientific Research from the Japanese Ministry of Education, Culture, Sports, Science and Technology is most gratefully acknowledged.

References

1. A.J. Waldorf, J.A. Dobrowolski, B.T. Sullivan, L.M. Plante, *Appl. Opt.* **32**, 5583 (1993)
2. M. Gilo, N. Croitoru, *Thin Solid Films* **350**, 203 (1999)
3. J. Aarik, H. Mandar, M. Kirm, L. Pung, *Thin Solid Films* **466**, 41 (2004)
4. G.D. Wilk, R.M. Wallace, J.M. Anthony, *J. Appl. Phys.* **89**, 5243 (2001)
5. S. Sayan, S. Aravamudhan, B.W. Busch, W.H. Schulte, F. Cosandey, G.D. Wilk, T. Gustafsson, E. Garfunfel, *J. Vac. Sci. Technol. A* **20**, 507 (2002)
6. J. Tsuaur, K. Onodera, T. Kobayashi, Z.J. Wang, S. Heisig, R. Maeda, T. Suga, *Sens. Actuators, A, Phys.* (2005) in press
7. A. Callegari, E. Cartier, M. Gribelyuk, H.F. Okorn-Schmidt, T. Zabel, *J. Appl. Phys.* **90**, 6466 (2001)
8. T. Nishide, S. Honda, M. Matsuura, M. Ide, *Thin Solid Films* **371**, 61 (2000)
9. K. Kukli, M. Ritala, M. Leskela, T. Sajavaara, J. Keinonen, A.C. Jones, J.L. Roberts, *Chem. Vap. Depos.* **9**, 315 (2003)
10. A. Abrutis, L.G. Hubert-Pfalzgraf, S. V. Pasko, A. Bartasyte, F. Weiss, V. Janickis, *J. Cryst. Growth* **267**, 539 (2004)
11. J. D. Wright, N.A.J.M. Sommerdijk, *Sol-Gel Materials Chemistry and Applications* (Gordon and Breach Science Publisher, Netherlands, 2001)
12. JCPDS Card No. 6-0318, Joint Committee for Powder Diffraction Standards, Swarthmore, Pennsylvania, 1984
13. M. Ritala, M. Leskela, L. Niinisto, T. Prohaska, G. Friedbacher, M. Grasserbauer, *Thin Solid Films* **250**, 72 (1994)
14. J. Aarik, A. Aidla, A.-A. Kuusler, T. Uustare, V. Sammelselg, *Thin Solid Films*, **340**, 110 (1999)
15. M.-Y. Ho, H. Gong, G.D. Wilk, B.W. Busch, M.L. Green, P.M. Voyles, D.A. Muller, M. Bude, W.H. Lin, A. See, M.E. Loomans, S.K. Lahiri, P.I. Raisenen, *J. Appl. Phys.* **93**, 1477 (2003)
16. S.-W. Nam, J.-H. Yoo, S. Nam, H.-J. Choi, D. Lee, D.H. Ko, J.H. Moon, J.-H. Ku, S. Choi, *J. Non-Cryst. Solids* **303**, 139 (2002)
17. D.P. Norton, *Mater. Sci. Eng.* **R43**, 139 (2004)